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Stability of CdS-coated TiO₂ solar cells

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Abstract The obstacle to realize the large-scale production of dye-sensitized solar cells (DSSCs) is its long-term stability and reliability problem. One of the main causes of the instability of DSSCs is the use of liquid electrolytes. In addition, exploring nano-sized particles of CdS as an alternative sensitizer for organic dye in dye-sensitized solar cells have attracted great interest due to the high cost and the instability of the organic dye. Our study has found that the CdS-coated TiO₂ cell degrades rapidly in the liquid electrolytes even under dark environment. In this work, a solid-state solar cell structure of Glass/FTO/TiO2/CdS:Cu/ FTO/glass was successfully made with an efficiency of 0.7%. CdS:Cu served as both the p-type conductor and absorber. No efficiency was obtained for cell structures of glass/FTO/TiO2/CdS/FTO/glass. This indicates the effectiveness of hole conducting behavior of CdS:Cu. This is the first time that this type of solid-state solar cell is reported and improved stability is demonstrated.

Keywords Dye-sensitized solar cell · Solid state · CdS · Stability

Introduction

Dye-sensitized solar cell (DSSC) has been widely studied in the past 10 years because of its low cost and potential high energy conversion efficiency. The highest energy conversion efficiency obtained is 11.5% (2009) under AM

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Department of Chemical & Paper Engineering, Miami University, Engineering Building, 64K, Oxford, OH 45056, USA e-mail: kerrll@muohio.edu 1.5 G simulated sunlight made from mesoporous TiO₂ films absorbed ruthenium dye and I_3/I^- liquid electrolytes [1]. However, the use of organic dye and liquid electrolytes has been the grand challenge in the long-term stability of the DSSC technology. A few solid-state inorganic p-type semiconductors were used as hole collectors to replace liquid electrolytes, such as CuI [2, 3], CuBr [4], CuSCN [5, 6], NiO [7] and CuAlO₂ [7], but their efficiency is very poor despite their good pore filling. The main reason is that the conductivity of these p-type conductors is low [8-10]. p-type Cu-doped CdS was made with simple chemical bath method with good conductivity (2 S/cm) in 1993 [11] and has been demonstrated on the homojunction CdS/CdS:Cu diode. CdS:Cu is a better p-type conductor than CuI ($\sigma_{\rm H}$ ~ 0.01 S cm⁻¹), and its crystal size can be as small as nanosize which will provide good pore filling. In addition, semiconductors such as CdS [12-14], CdSe [15], PbS [16] have been used to replace the expensive organic dyes as alternative sensitizers. CdS is one of the most commonly used alternative sensitizers due to its several advantages. For example, its optical properties and the band gap can be adjusted by the particle size [17]; it has better UV stability than organic dyes [18] and it has reduced dark current [19]. However, the current efficiency of CdS-sensitized solar cell is still low. The highest efficiency of CdS-sensitized TiO₂ solar cell is 1.84% [17, 20, 21] and the cell is very unstable. In addition, the authors did not report the efficiency of any control sample. The CdS-sensitized cell efficiency can be improved to 2.8% [22] by adding CdSe. However, adding CdSe complicates the sensitizing process. The poor cell efficiency and its instability in I_3^{-}/Γ electrolytes have limited the application of CdS in DSSC solar cells. Researchers have attributed the low efficiency to the poor pore penetration of CdS into TiO_2 or ZnO network [23, 24]. In this work, a CdS process is developed and achieved a solar cell efficiency of 2.46% (V_{oc} =0.69 V, J_{sc} =6.54 mA/cm², F.F.=0.55). This is the highest reported efficiency of using CdS alone as sensitizer (active area of 0.16 cm^2 under AM 1.5). This process can be easily modified to achieve p-type Cu-doped CdS (CdS:Cu) which allows the manufacturing of a simple solid-state CdScoated nanocrystalline TiO2 solar cells. A solid-state solar cell structure of Glass/FTO/TiO2/CdS:Cu/FTO/glass was successfully made with an efficiency of 0.7%. Compared to the solid-state DSSCs made from CuI [2, 3], CuBr [4], CuSCN [5, 6], NiO [7], and CuAlO₂ [7], this type of solar cell has very simple controllable structure and does not use the expensive dye and Pt catalyst. The CdS:Cu serves as both the absorber and hole conductor. Even though the efficiency of this type of solar cell is still lower than that of CuI, CuBr, and CuSCN and further optimization of CdS:Cu is needed to improve the cell performance, the demonstrated stability of this type of solar cell indicates that this novel cell structure and new hole conductor (CdS:Cu) could be promising approaches to achieve stable solid-state DSSCs.

Experimental

TiO₂ electrode

Commercial TiO₂ (3 g; P₂₅, Degussa AG, Germany) was mixed with acetylacetone, Triton X-100 and water [25]; 10% PEG 20,000 (by weight) was subsequently added in the paste. TiO₂ pastes were then spread on SnO₂:F-coated (FTO) glass substrate using a doctor blade method. The film was annealed in 450 °C for 30 min in air. The thickness of TiO₂ is about 10 μ m.

CdS deposition

TiO₂-coated FTO was put into the solution of 0.015 M CdCl₂ 50 ml, 1.5 M thiourea 25 ml, and ammonium hydroxide 62.5 ml mixed in 366 ml water. The solution pH was around 11.5 [26]. The reaction was carried out at 60°C for 20 min with slow stirring. The as-grown sample exhibited nanoparticle structure with particle size of 100 nm (Fig. 1).

Solar cell fabrication

Electrolyte was made with 0.1 M LiI (Aldrich), 0.05 M I₂ (Aldrich), 0.6 M tetrabutylammonium iodide (Fluka), and 0.5 M *tert*-butylpyridine (Aldrich) in dry acetonitrile (Fluka) [27]. The counter electrode was made of Pt-coated FTO glasses [28]. The active area of the cell is 0.16 cm². AM 1.5 solar simulator (100 mW·cm⁻²) was used as illumination source and *I–V* curves were obtained with a digital source meter (Keithley 2,400, Keithley Instruments). The measurement is calibrated with commercial Si refer-



Fig. 1 SEM images of CdS on glass

ence solar cell. Two configurations of solar cell structure are investigated in this study. One is the CdS-coated TiO_2 solar cell with liquid electrolytes (Fig. 2a). Another is CdS-coated TiO_2 solid-state solar cells (Fig. 2b). The solid-state solar cell does not require the use of Pt catalyst since the liquid electrolyte is not employed. A control sample made with the standard Ru-dye is shown in Fig. 2c.

Results and discussion

CdS morphology and CdS penetration into TiO₂ network

Figure 3 shows the cross section of the CdS/TiO₂/FTO/ glass. The energy-dispersive X-ray spectroscopy (EDS) was



Fig. 2 Three solar cell configurations investigated in this work a CdS-coated TiO_2 solar cell with liquid electroytes b Solid-state CdS-coated TiO_2 solar cell c Control sample of Ru-dye-sensitized TiO_2 solar cell with liquid electrolytes

Fig. 3 Cross section of CdS on TiO_2 and the EDS testing points



taken for locations A, B, and C along the cross section as shown in Fig. 3. As we can see, the sulfur ion has better penetration into the TiO₂ film than Cd ion which might be due to the relative larger particle size of CdS. This suggests that further reduction in CdS particle size is required to improve the pore penetration. Nevertheless, the efficiency of the solar cell obtained by this CdS with liquid electrolytes (structure shown in Fig. 2a) is 2.46% $(V_{\rm oc}=0.69 \text{ V}, J_{\rm sc}=6.54 \text{ mA/cm}^2, \text{ F.F.}=0.55)$. This is the highest reported efficiency of using CdS alone as sensitizer. Compared to the control sample (structure shown in Fig. 2c) efficiency of 7.88% (V_{oc} =0.73 V, J_{sc} = 19.22 mA/cm², and F.F.=0.56), the low J_{sc} is the major factor contributing to the poor cell performance which might be due to the poor CdS penetration into TiO₂ film and inefficient light absorption of CdS compared to the dye.

Investigation of CdS-coated solar cell stability

In order to test the stability of CdS-coated solar cell, solar cells with liquid electrolytes (structure shown in Fig. 2a) were sealed and kept in the dark for 6 h. The cells suffered significant J_{sc} degradation as seen in Table 1 (the first two rows of "w/o barrier"). To explain this, we need to understand the chemical reactions among the three compo-

nents of CdS, TiO₂, and liquid electrolytes. Samples of FTO/CdS, FTO/TiO₂ and FTO/TiO₂/CdS were immersed into 5 mL electrolytes (1/25 diluted with acetonitrile) for 6 h in the dark. UV–Vis data of the electrolyte solution shown in Fig. 4 was measured before and after the dipping of samples. FTO/TiO₂ and FTO/CdS do not show any UV–Vis degradation (overlapping of UV–Vis curves with that of original electrolytes) in electrolytes after 6 h which means that the reaction can be neglected when only having TiO₂ or CdS alone on FTO in electrolytes. FTO/TiO₂/CdS (the top dashed line in Fig. 4), on the other hand, showed the dramatic change in UV–Vis. This indicates that TiO₂ may serve as an oxidizer for CdS and accelerate the reaction between CdS and electrolytes. Thus, a compact barrier layer of TiO₂ film less than 2 μ m (Fig. 5a) is

Table 1 Stability study of CdS-sensitized TiO_2 liquid electrolytes solar cells with (W/) and without (W/O) TiO_2 barrier layer

TiO ₂ film	Time (h)	Efficiency (%)	$V_{\rm oc}$ (V)	$J_{\rm sc}~({\rm mA/cm}^{\ 2})$	F.F.
W/O barrier	0	2.46	0.69	6.54	0.55
W/O barrier	6	0.52	0.81	3.82	0.17
W/barrier	0	2.41	0.67	6.79	0.53
W/barrier	6	1.09	0.85	3.05	0.43





deposited on porous TiO₂ to have structure of TiO₂/CdS/ TiO₂. To make TiO₂ barrier film, 3 g commercial TiO₂ (P₂₅, Degussa AG, Germany) was ground in a porcelain mortar with a small amount of water 1 mL containing acetylacetone 0.1 mL (about two drops) to prevent reaggregation of the particles. After the powder had been dispersed by the high shear forces in the viscous paste, it was diluted by slow addition of water (4 mL) under continued grinding. When TiO_2 barrier film was applied, the J_{sc} degradation slowed down as seen in Table 1 (the last two rows of "w/barrier"). The TiO₂ barrier film does not block hole transport from electrolytes to CdS possibly



Fig. 5 SEM image of TiO_2 barrier film of $TiO_2/CdS/TiO_2$ and band diagram of different layers in the solar cell [29]



Fig. 6 The absorption spectrum of $\rm TiO_2$ before and after CdS and CdS:Cu deposition

because the $E_{\rm vb}$ of TiO₂ (3.0 NHE) is lower than CdS (2.0 NHE) [29] (Fig. 5b). The solar cell performance data in Table 1 has demonstrated the modest effectiveness of the TiO₂ barrier layer.

Solid-state CdS-coated solar cell

Using TiO₂ barrier layer only demonstrated modest cell stability. Since the accelerated reaction between TiO₂ and CdS occurs at the presence of liquid electrolytes, a solid-state CdS-coated TiO₂ solar cells (structure shown in Fig. 2b) is attempted. To make CdS:Cu on TiO₂, TiO₂-coated FTO was immersed in the solution of 0.015 M CdCl₂ 50 ml, 1.5 M thiourea 25 ml and ammonium hydroxide 62.5 ml mixed in 366 ml water. 5 mol% CuCl was added as Cu source. The pH of the solution was around 11.5. The reaction was carried out at 60 °C for 20 min and with slow stirring.

Solid-state solar cell structure using CdS:Cu as both sensitizer and hole conductor was obtained with cell efficiency of 0.7% (J_{sc} =2.04 mA/cm², V_{oc} =0.68 V and F.F.=0.51). The *I–V* curve is shown in Fig. 8. A thin layer of graphite is used as the contact between CdS:Cu and FTO. This is the first time this type of solid-state TiO₂ nanocrystalline solar cell is made. No cell efficiency was found for cells with FTO/TiO₂/CdS/



Fig. 7 a SEM plane view of CdS:Cu and b EDS of CdS:Cu

Fig. 8 *I–V* curves of the solid-state CdS:Cu devices as assembled (0.7%; J_{sc} =2.04 mA/cm², V_{oc} =0.68 V, and F.F.=0.51) and after 20 h (0.68% with J_{sc} = 2.34 mA/cm², V_{oc} =0.68 V, and F. F.=0.42)



FTO structure. This indicates the hole conducting behavior of CdS:Cu. The absorption spectrum of TiO₂ before and after CdS and CdS:Cu deposition is presented in Fig. 6. The decreased light absorption of CdS:Cu could be caused by the increased particle size of CdS:Cu (SEM shown in Fig. 7a) compared to CdS (SEM shown in Fig. 1). The EDS data shown in Fig. 7b indicates that the Cu dopant concentration is about 3 at% in CdS:Cu film. Thus, if the light absorption of CdS:Cu could be improved by reducing the CdS:Cu particle size, the efficiency of solid-state CdS-coated TiO₂ solar cell will be increased. This solid-state CdS-coated TiO₂ solar cell has demonstrated much better stability for days. The cell efficiency is 0.68% with $J_{sc}=2.34 \text{ mA/cm}^2$, $V_{oc}=0.68 \text{ V}$, and F.F.=0.42 after 20 h as shown in Fig. 8. Further optimization is needed to improve the $J_{\rm sc}$ by optimizing the CdS:Cu deposition process to reduce the particle size of CdS:Cu.

Conclusions

In this study, a solar cell efficiency of 2.46% was achieved for CdS-coated TiO₂ solar cell. However, the cell was found to be unstable due to the reaction with electrolytes. Adding TiO₂ barrier layer can improve the cell stability to some extent. A complete solid-state CdS-coated solar cell using CdS:Cu as both hole conductor and sensitizer was made, improving cell stability dramatically. The solid-state solar cell efficiency was 0.7%. This is the first time this type of solid-state TiO₂ nanocrystalline solar cell is made.

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